Nanostructures in p-GaAs with improved tunability

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A nanofabrication technique is presented which enables the fabrication of highly tunable devices on p-type, C-doped GaAs/AlGaAs heterostructures containing shallow two-dimensional hole systems. The high tunability of these structures is provided by the complementary electrostatic effects of intrinsic in-plane gates and evaporated metallic top-gates. Quantum point contacts fabricated with this technique were tested by electrical conductance spectroscopy. © 2010 American Institute of Physics. [doi:10.1063/1.3463465]

The recently emerging interest in p-type nanodevices is twofold. On one hand the pronounced spin-orbit as well as carrier-carrier Coulomb interactions make these systems suitable for studying many-body phenomena in reduced dimensions. On the other hand, low-dimensional hole-doped systems represent promising candidates for the practical implementation of spin-based quantum information processing technology.¹ Hyperfine interactions, identified as one of the main sources of spin dephasing in n-type systems,² are expected to be strongly suppressed due to the p-type nature of the valence band holes.³ Additionally, while strong spin-orbit interactions lead to fast spin relaxation in higher dimensions, further confinement of holes into quantum dots can significantly increase the relaxation time of the holes' spins,⁴ so that it can be comparable, or even larger than the one of the electron spins as it was tested by optical pumping experiments in self-assembled p-type quantum dots.^{3,5}

In spite of the above prospects, low-dimensional hole systems have received less attention due to technological difficulties in fabricating stable p-type structures. Nanodevices defined on shallow two-dimensional hole gases (2DHGs) in p-type GaAs by conventional split-gate techniques show significant gate instabilities. This is presumably due to the fact that metallic Schottky barriers on p-type GaAs are leaky and exhibit a highly hysteretic behavior, making them unsuitable for high-precision tuning of the devices. In previous works^{6–8} local anodic oxidation lithography⁹ carried out by a charged tip of an atomic force microscope (AFM) has been utilized as an alternative method for establishing stable nanoconstrictions tuned by intrinsic in-plane side-gates.¹⁰

However, the tunability of the nanostructures patterned on p-GaAs turns out to be much more sensitive to specific oxide line parameters compared to those implemented in n-GaAs resulting in a lower yield of the functional devices. In this paper we demonstrate that combining the electrostatic effect of such in-plane gates with the one of an evaporated metallic top-gate considerably increases the tunability of the p-type structures while the stability of the top-gate can be optimized by the proper choice of a gate insulator oxide. We investigated several quantum point contacts (QPCs)^{11–13} as test devices patterned by AFM lithography or, alternatively,

by electron beam lithography (EBL) and wet chemical etching prior to top-gate deposition. All QPCs were characterized at temperatures between 5 K and 100 mK by four-terminal lock-in measurements of the linear conductance G at a frequency of 31 Hz. Here we present representative data obtained on an EBL defined QPC.

Previous studies^{14–18} focused mostly on devices fabricated on (311) oriented deep 2DHGs in Si-doped GaAs, exhibiting an anisotropic in-plane Fermi surface. Here we present data on C-doped (100) samples containing a shallow 2DHG with an in-plane isotropic electronic structure. In contrast to similar quality devices implemented in (311) oriented 2DHGs the functionality of devices patterned on C-doped GaAs wafers as well as the interpretation of the transport experiments in these devices are expected to be independent of the particular orientation of the device with respect to the crystal orientation. The QPCs were oriented along the high symmetry crystallographic directions parallel to the cleaved edges of the wafer. The host heterostructure consists of a 5 nm undoped GaAs cap layer, followed by a 15 nm thick, homogeneously doped layer of AlGaAs separated from the 2DHG by a 25 nm thick, undoped AlGaAs spacer layer.¹⁹

A 20 nm thick HfO₂ layer deposited as a gate insulator between the GaAs wafer and the metallic top-gate was grown by atomic layer deposition (ALD) in a Picosun Sunale R-150B system. A linear growth rate of 0.08 nm per pulse cycle has been found at a reaction chamber temperature of 250 °C and at a tetrakis(ethylmethylamido)hafnium source temperature of 90 °C. The structural quality of the resulting polycrystalline HfO₂ layer have been investigated by scanning electron microscopy (SEM) on freshly cleaved GaAs wafers as illustrated in Fig. 1(a). The surface roughness determined by AFM is found to be 4 nm rms, independent of the layer thickness.

The 10 nm thick Ti sticking layer and the 90 nm thick Au layer of the metallic top-gates were deposited by electron beam evaporation in a Leybold Univex 500 chamber at a base pressure of 10^{−7} mbar. The electronic properties of the oxide layer as well as the tunability of the host 2DHG by the top-gate were tested separately at $T=4.2$ K on a 100 μm wide Hall-bar device defined by conventional optical lithography. The breakdown characteristics of the 20 nm thick oxide layer shows no trace of leakage currents in the -8 V $< V_{ig} < 8$ V top-gate voltage regime and is reproducible as long as the high voltage leakage current is kept below 0.5 nA

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FIG. 1. (a) SEM image of a freshly cleaved GaAs wafer covered by a 20 nm thick HfO_2 layer. (b) Breakdown characteristics of the top gate at $T=4.2$ K. (c) Hole mobility μ as a function of the density P in the 2DHG in the range of $-8 \text{ V} < V_{\text{tg}} < 8 \text{ V}$ at $T=4.2$ K.

as shown in Fig. 1(b). The hole density P and mobility μ of the 2DHG were determined by conventional four-terminal longitudinal and Hall resistance measurements performed at low magnetic fields up to 0.1 T. The tunability of P in the 2DHG is, however, limited to about 20%–25% within the leakage free top-gate voltage regime [Fig. 1(c)]. This is in contrast to the estimated depletion voltage of $V_{\text{tg}} \approx 0.5 \text{ V}$ based on a plate capacitor model taking the thicknesses and the dielectric constants of the layers into account. This is presumably due to the screening effect of the low mobility charges residing in the doping layer and at the interface between the insulator and the semiconductor. The mobility edge arising from the background potential roughness of the 2DHG is close to the density of the ungated sample $P \approx 4 \times 10^{11} \text{ cm}^{-2}$.

The structural and electronic properties of the etched lines separating the in-plane gates were individually characterized prior to top-gate deposition. Line widths of 140–170 nm and depths of 15–20 nm provided breakdown voltages on the in-plane gates in the range of a few 100 mV as illustrated in Fig. 2(b) for the EBL defined QPC shown in Fig. 2(a).

Figure 3(a) shows the absolute value of the leakage current across a separating line in the leakage free gate configuration regime defined as $I_{\text{leak}} < 0.1 \text{ nA}$ when both the V_{tg} top-gate and one of the in-plane gate voltages [shown for V_{G1} in Fig. 3(a)] are varied. The overall broadening of the leakage free domain in V_{G1} with respect to the $I_{\text{leak}}(V_{\text{G1}})$ trace recorded in the absence of the top-gate [Fig. 2(b)] is explained by the screening effect of the top-gate. The further broadening toward higher V_{tg} reveals the variation of the

FIG. 2. (a) AFM micrograph of the 200 nm wide QPC prior to top-gate deposition. The 2DHG is depleted under the etched lines defined by electron-beam lithography. The electronic width of the QPC is controlled by the G_1 and G_2 in-plane side gates. (b) Representative breakdown characteristics of the etched lines at $T=4.2$ K shown for gate G_1 .

FIG. 3. (Color online) (a) Leakage current I_{leak} across the in-plane barrier of gate G_1 as a function of the V_{tg} top-gate voltage and the V_{G1} in-plane gate voltage at $T=100$ mK. The colored area shows the magnitude of I_{leak} in the leakage free domain defined as $I_{\text{leak}} < 0.1 \text{ nA}$. (b) Color map of the QPC conductance G as a function of the V_{G1} and V_{G2} in-plane gate voltages at a representative top-gate voltage value of $V_{\text{tg}}=0.5 \text{ V}$. The white arrows indicate the V_g and ΔV symmetric and antisymmetric in-plane gate voltage configurations, respectively.

hole density in the 2DHG, i.e., a decreasing electrochemical potential with respect to the in-plane barriers. The asymmetry in the position of the border lines [also visible in Fig. 2(b)] reflects the asymmetry in the applied in-plane gate biases as one side of the separating lines was always kept at ground. The asymmetry in the curvature of the border lines is attributed to the nonlinearities in the tunneling probability across the barrier under asymmetric in-plane gate biases.

Figure 3(b) displays the conductance of the QPC shown in Fig. 2(a) as its electronic width is tuned by the G_1 and G_2 in-plane side gates at a fixed top-gate voltage value of $V_{\text{tg}}=0.5 \text{ V}$. The linear combination $V_g = \alpha_1 V_{\text{G1}} + \alpha_2 V_{\text{G2}}$ tunes the confining potential in a symmetric fashion, where $-\alpha_1/\alpha_2$ is the slope of the lines of equal linear conductance on the $V_{\text{G1}}-V_{\text{G2}}$ plane. Accordingly, the asymmetric gate voltage combination $\Delta V = \alpha_2 V_{\text{G1}} - \alpha_1 V_{\text{G2}}$ leads to a transverse electric field resulting in a lateral shift in the QPC axis while the Fermi energy is kept constant by the leads.

In a simple model of the QPC potential¹³ V_{tg} controls the position of the electrochemical potential with respect to the QPC barrier, while V_g is expected to set the electronic width of the QPC confinement and thus, also the separation of the one-dimensional subbands. In this picture the onset of the finite conductance occurs as the electrochemical potential approaches the first QPC subband. An increased tunability of the QPCs is expected by having the individual control over both parameters.

The stability of the conductance traces upon different top-gate sweeps is illustrated in Fig. 4(a). Sweeping V_{tg} over the extended regime of Fig. 3(a) at a fixed V_g value results in a hysteretic behavior due to the charging of the defect sites in the doping as well as in the oxide layer. On the other hand, a top-gate sweep performed in the limited voltage range displayed in the inset of Fig. 4(a) yields to hysteresis-free, stable conductance traces. The tunability of the QPC by the in-plane gates at various V_{tg} values is shown in Fig. 4(c). It demonstrates that while the overall shape of the individual conductance curves is slightly affected, the range of tunability by means of V_g can be optimized by the proper choice of the additional control parameter V_{tg} . Moreover, our experience shows that QPCs which were not tunable prior to top-gate deposition due to a large misalignment between the

FIG. 4. (Color online) (a) QPC conductance G as a function of V_{tg} at $T=100$ mK while V_g is kept at a constant value of 0.5 V. The arrows indicate the direction of the consecutive V_{tg} sweeps. The inset shows a hysteresis-free top-gate sweep in a limited V_{tg} regime. (b) G as a function of V_g at various temperatures between $T=5$ K and 100 mK at $V_{tg}=1.4$ V. (c) G as a function of V_g recorded at different top-gate voltages ranging from $V_{tg}=2$ V (leftmost trace) to $V_{tg}=0.6$ V (rightmost trace).

electrochemical potential and the first QPC subband become tunable by changing V_{tg} .

At more positive V_{tg} the density in the leads is lower and the opening of the QPC requires more negative voltages on the in-plane gates corresponding to a broader lateral confinement. At the same time, the length of the observed quantized plateaus slightly increase in agreement with the saddle-point potential model.¹³ Based on the temperature dependence shown in Fig. 4(b) along with further criteria for the associated zero bias anomaly discussed elsewhere²⁰ we attribute the resonant feature on the rise of the $2e^2/h$ plateau to a transmission resonance arising from the interaction with an impurity state inside the QPC. The effect of this impurity resonance on the transmission becomes more pronounced toward narrower QPC confinements set by less positive V_{tg} top-gate voltages. It has to be added that the majority of the hole QPCs exhibit a different conductance feature below the $2e^2/h$ plateau with a markedly different temperature dependence which we attribute to the so-called 0.7 anomaly,¹⁸ as reported in Ref. 20

In conclusion, we established a hybrid fabrication method for nanostructures implemented in shallow 2DHGs of GaAs heterostructures utilizing the complementary electrostatic effects of intrinsic in-plane gates and evaporated metallic top-gates. The electric properties of the in-plane bar-

riers as well as of the HfO_2 top-gate insulator layer were characterized separately. The increased tunability of QPCs fabricated by this method was demonstrated by electrical conductance spectroscopy.

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