Terahertz-Driven Phonon Upconversion in SrTiO₃

M. Kozina,¹ M. Fechner,² P. Marsik,³ T. van Driel,¹ J.M. Glowina,¹ C. Bernhard,³ M. Radovic,⁴ D.

Zhu,¹ S. Bonetti,⁵ U. Staub,⁴ and M.C. Hoffmann¹

¹*Linac Coherent Light Source, SLAC National Accelerator Laboratory, Menlo Park, California 94025, USA*

²*Max Planck Institute for the Structure and Dynamics of Matter, 22761 Hamburg, Germany*

³*Department of Physics, University of Fribourg, Chemin du Musée 3, CH-1700 Fribourg, Switzerland*

⁴*Swiss Light Source, Paul Scherrer Institut, 5232 Villigen PSI, Switzerland*

⁵*Department of Physics, Stockholm University, SE-106 91 Stockholm, Sweden*

Abstract

Direct manipulation of the atomic lattice using intense long-wavelength laser pulses has become a viable approach to create new states of matter in complex materials. Conventionally, a high frequency vibrational mode is driven resonantly by a mid-infrared laser pulse and the lattice structure is modified through indirect coupling of this infrared-active phonon to other, lower frequency lattice modulations. Here, we drive the *lowest* frequency optical phonon in the prototypical transition metal oxide SrTiO₃ well into the anharmonic regime with an intense terahertz field. We show that it is possible to transfer energy to *higher* frequency phonon modes through nonlinear coupling. Our observations are carried out by directly mapping the lattice response to the coherent drive field with femtosecond x-ray pulses, enabling direct visualization of the atomic displacements.

In many complex condensed matter systems, small changes to the crystal lattice structure drastically alter electronic properties including conductivity, polarization, orbital, charge, and spin-order¹⁻⁵. This closeness of states results in rich and involved phase diagrams, whose exploration conventionally requires changing external parameters such as chemical composition, temperature, static pressure, strain or magnetic fields⁵⁻⁸. Additionally, optical excitation using ultrashort laser pulses has been used to dynamically change the electronic and crystal structure^{1-4,9}. This method allows access to non-equilibrium states of matter that show unconventional and unique properties. Recent developments in mid-infrared laser sources have enabled a new route to control material properties: resonant excitation of phonon modes to dynamically alter the lattice structure and phonon-phonon coupling are exploited to coherently control inaccessible (i.e. non-infrared-active) phonon modes¹⁰. This approach is often dubbed “*nonlinear phononics*” and requires precisely tuned mid-infrared radiation and impulsive excitation¹¹.

Here we show a novel, direct approach to alter the lattice structure by driving the lowest frequency optical phonon (the soft mode) of the incipient ferroelectric SrTiO₃ (STO) into the strongly nonlinear regime with intense broadband terahertz (THz) radiation and by observing with ultrafast x-ray diffraction. Specifically, we demonstrate that it is possible to transfer energy from a *low-frequency* phonon mode to *higher-frequency* modes, recently theoretically predicted¹², and to detect these changes directly in the structure. We anticipate that our approach will open up a deeper understanding of quantum materials through direct manipulation of the lattice structure and the utilization of dynamic coupling of vibrational modes. For example, control of the lattice structure of a substrate material such as STO can be used for indirect control of thin film properties through strain engineering¹³. Indirect optical phonon

excitation could also impact thin-film and interfacial superconductivity in STO systems where electron-phonon coupling may play an important role^{14,15}. Moreover, driving the STO soft mode far into the nonlinear regime has potential impact on the interplay between superconductivity and ferroelectricity near the quantum critical point^{16,17}.

STO has a low-frequency zone-center transverse optical (TO) phonon that is infrared (IR) active and highly temperature dependent^{18–20}. This so called soft-mode phonon mediates the ferroelectric phase transition in primarily displacive ferroelectrics such as BaTiO₃ or PbTiO₃^{21–23}. In bulk STO, while the soft mode shifts to lower frequencies with decreasing temperature, the system never reaches a ferroelectric state due to quantum fluctuations²⁴. The temperature dependence and large anharmonicity make this mode ideally suited for probing nonlinear phonon interactions. While all-optical techniques can yield indirect evidence of excited phonons^{25–27}, x-ray diffraction measurements of the structural evolution of the STO unit cell provide quantitative information about the atomic motion. Recent developments in hard x-ray free electron lasers enable these ultrafast structural determinations^{28–30}.

We illustrate the principle of our experiment in Fig. 1. A strong-field THz pulse containing spectral components from approximately 0.2 to 2.5 THz interacts with the soft mode phonon of an STO thin film (50 nm). The frequency of the soft mode phonon can be tuned in the range of 1.5 to 2.5 THz by varying the sample temperature³¹. The relative positions of the atoms in the unit cell (Sr, turquoise; Ti, red; O, white) is monitored by diffraction of femtosecond x-ray pulses with duration (~30 fs) considerably shorter than a cycle of the THz field (see Methods below).

Figure 1: Phonon frequency spectrum and experimental overview. **a**, Strong THz radiation (red) interacts with the STO soft mode phonon (yellow). The degree of resonant overlap can be tuned by temperature. Energy is exchanged with higher frequency phonon modes (turquoise, purple) through nonlinear coupling. The STO unit cell and two lowest-frequency zone-center TO eigenmodes are indicated at the top of the figure. **b**, Phonon motion is probed in the time domain with ultrafast x-ray diffraction in reflection geometry.

Figure 2: Time-resolved x-ray diffraction data at 100 K. **a**, Intensity change of the (2 -2 3) peak in STO (black) as function of time delay compared to the THz pump field (red). The traces are shifted to overlap in time as their relative timing is only known within 1 ps. **b**, Fourier transform of the time domain x-ray data shows distinct peaks, identified as the soft mode at 1.5 THz, the **TO₂** mode at 5.15 THz and the **TO₃** mode at 7.6 THz. The spectrum of the THz excitation pulse is shown in red. **c**, A short-time Fourier transformation with a 2.5 ps Kaiser window reveals the immediate onset of oscillations. The red shaded area indicates the spectrogram of the THz pump pulse (contour at 30% of peak magnitude).

Figure 2a shows the time-resolved change in the x-ray scattering (trXRD) intensity of the (2 -2 3) STO peak at ~100 K (black), overlaid with the electric field of the incident THz radiation (E_{THz} , red). The trXRD and E_{THz} signals are shown with their peaks overlapped in time since their relative time of arrival is only known with a precision of about 1 ps. At early times (<1.5 ps) the trXRD response appears to closely follow E_{THz} but then shows strong saturation and a number of additional high frequency features after the main peak which do not appear in E_{THz} . The spectral content of both data sets is shown in Fig. 2b normalized to a peak value of unity. At low frequencies (< 1 THz), the spectral content of E_{THz} and trXRD are very similar, suggestive that the motion of the ions in the STO film primarily follows the driving THz field ³². However, in the frequency band between one and two THz (shaded in yellow), the trXRD signal has a much larger magnitude than E_{THz} . Moreover, while the driving field spectral content is essentially zero at frequencies higher than 3 THz, there are clear features in the trXRD signal at 5.15 THz and 7.6 THz (shaded turquoise and purple, respectively). We interpret the spectral weight in the 1-2 THz region as a signature of the resonant excitation of the soft mode, which is well known to have strong IR absorption at these frequencies below room temperature and is strongly damped^{33,34}. The soft mode can be excited directly, since there is still significant spectral overlap between the THz pump pulse and the absorption spectrum of the phonon³⁵.

We identify the features at 5.15 THz and 7.6 THz as contributions from two additional transverse optical phonon modes. Bulk STO at room temperature is known to have at zone-center transverse and longitudinal optical phonons in the range 5.1-5.3 THz, as well as a silent phonon at 7.95 THz^{34,36–38}. In our thin-film sample, the system becomes slightly tetragonal because of lattice matching to the substrate thus lifting the mode degeneracies; we expect to

couple only to the in-plane E-symmetry transverse optic modes parallel to the THz electric field (see supplement for details). We label the three modes (soft mode, 5.15 THz mode, and 7.6 THz mode) as TO_1 , TO_2 , and TO_3 . The higher frequency modes, in contrast to the soft mode, cannot be directly driven by the THz pulse because they lack spectral overlap. Notably, the 7.95 THz mode in bulk STO is a *silent* mode that has neither infrared nor Raman cross section and can only be observed through higher order spectroscopy methods such as hyper-Raman³⁸. While IR activity is no longer symmetry-forbidden for the 7.95 THz mode in our strained film, our ellipsometry measurements at room temperature reveal no signature of this (see supplement) and other measurements on a different STO/LSAT thin film system show no IR activity for this mode at either 5 K or 300 K³⁹. Instead, we propose that they are driven indirectly by coupling to the soft mode which is driven so far from equilibrium as to produce new frequency components because of its large anharmonicity. The (broad) third and fifth harmonics of the soft mode resonantly overlap with the TO_2 and TO_3 modes, enabling efficient energy transfer. These modes are much narrower than the soft mode and so we attribute the sharp peaks at 5.15 THz and 7.6 THz in the trXRD signal to their excitation rather than the broad soft-mode harmonics.

The dynamics of the excitation process can be illustrated by calculating the time-resolved change in spectral content. In Figure 2c we present a time-frequency plot of the x-ray data in Fig. 2a using a 2.5 ps full-width, half-maximum sliding window fast Fourier transform (FFT). The red shaded area is a contour (30% of peak magnitude) of the same analysis for the E_{THz} trace. While there is clear overlap at early times between the x-ray and THz responses, there are persistent frequency components in three bands: 1-2 THz (soft mode band), 4.5-5.5 THz (TO_2 band), and 7-

8 THz (TO_3 band). Within the resolution of the time window, both high frequency bands begin to oscillate at the same time, immediately after the main low frequency response.

To test our hypothesis of nonlinear coupling through the soft mode phonon, we tuned the soft mode phonon frequency in and out of resonance with the driving THz field by varying the sample temperature. The THz peak field strength was kept constant at 880 ± 50 kV/cm. Figure 3a shows the sample optical conductivity (measured by THz ellipsometry³¹) as a function of temperature overlaid with the spectrum of E_{THz} . The mode frequency changes from 1.5 THz at 100 K to >2.5 THz at 250 K and thus moves out of resonance with the spectral components of E_{THz} . In Fig. 3b we show the magnitude of the FFT of the trXRD signal as a function of temperature, normalized to the peak of the 100 K trace. As we move out of resonance with the THz pulse (increasing temperature), the spectral weight between 1-2 THz diminishes, consistent with our model of resonant excitation. Furthermore, the peaks at 5.15 THz and 7.6 THz disappear as the temperature is increased, because the soft mode amplitude is no longer large enough to couple to these modes.

Figure 3: Temperature dependence of sample response. *a*, Optical conductivity data measured by THz ellipsometry show the temperature dependence of the soft phonon mode. Only at the lowest temperatures is there

*significant overlap with the THz pump spectrum (red area, normalized units) to drive the soft mode into the nonlinear regime. **b**, Fourier transform of trXRD signal showing transfer to high frequency modes at the corresponding temperatures in **a**.*

Alternatively, we can tune the energy coupled to the soft mode by adjusting the THz field strength. Here we studied the sample response while the incident THz pulse was continuously attenuated with a pair of wiregrid polarizers without affecting the pulse shape or spectrum. The highest field value was achieved by removing the polarizers altogether. The trXRD signal with increasing THz field strength is shown in Fig. 4a. Several hallmarks of nonlinear phenomena are present: the trXRD peak signal saturates as a function of THz field strength, manifest as a sublinear relation between peak signal and applied field (Fig. 4b); the rise of the primary peak (near 1.8 ps) also grows steeper with increasing THz field, indicating new frequency content generated through a nonlinear structural response; and, at the very highest fields, we observe the onset of high frequency oscillations superimposed upon the primary signal. For clarity, in Fig. 4c we emphasize the high frequency response by passing a subset of the data in Fig. 4a through a 3.5 THz high-pass filter. We see a turn-on of the high frequency oscillations at approximately 1.5 ps, near the first minimum of the trXRD response, suggestive that the large amplitude motion of the soft mode enables energy transfer into the higher frequency components. In Fig. 4d we present the trXRD spectrum for the same peak THz fields as in Fig. 4c, indicating the scaling of the TO₂ zone center mode at 5.15 THz.

Figure 4: THz field dependent data and model calculations. **a**, Time-resolved change in x-ray scattering intensity for several THz pulse peak electric field values. Model data are shown as black solid lines. **b**, Maximum x-ray intensity change at increasing THz peak field strength (black dots) illustrating the saturation effect due to the quartic nature of the potential. The solid line is our model. **c,d**, High frequency portion (>3.5 THz) of our data illustrating the onset of the TO_2 phonon oscillations in the time domain (**c**) and frequency domain (**d**). **e**, Phonon potential for soft mode used for simulations in **a** (blue). The gray shading represents the trajectory of the simulated soft mode at largest applied field (880 ± 50 kV/cm).

In the following, we develop a model based on a driven anharmonic oscillator with nonlinear coupling to higher frequency oscillators known from spectroscopy of bulk STO^{37,38}. We only consider coupling to the 5.15 THz mode to facilitate quantitative fits to our datasets for varying the THz field. We nevertheless point out that the appearance of the 7.6 THz mode is experimentally robust and our model could easily be extended to include this. The THz field will directly excite the soft mode phonon in-plane, along the polarization direction of the field (parallel to $[1 -1 0]$). We expect the modes to couple along the same displacement coordinate, and thus we simplify the mode displacements as scalar amplitudes along the THz field

polarization. We label the soft mode and TO₂ mode amplitudes respectively as Q_1 and Q_2 , and expand the lattice potential as follows:

$$V = \frac{1}{2}\omega_1^2 Q_1^2 + \frac{1}{2}\omega_2^2 Q_2^2 + \frac{1}{4}\kappa_1 Q_1^4 + \frac{1}{4}\kappa_2 Q_2^4 + \chi Q_1^2 Q_2^2 + \psi_{12} Q_1^3 Q_2 + \psi_{21} Q_1 Q_2^3. \quad (1)$$

Here, ω_i denote the eigenfrequencies and κ_i, ψ_{ij} and χ are the anharmonic phonon and phonon-phonon contributions to the potential. By symmetry only even terms are included in the potential. The time dependent dynamics are obtained from equation (1) by considering the equations of motion as:

$$\ddot{Q}_i + \frac{dV}{dQ_i} + \gamma_i \dot{Q}_i = Z_i^* \tilde{E}_{\text{THz}}. \quad (2)$$

We include additional terms containing γ_i to account for the lifetime of phonons and $Z_i^* \tilde{E}_{\text{THz}}$ to incorporate coupling to the driving field $\tilde{E}_{\text{THz}}$ via the mode effective charge Z_i^* ⁴⁰ and adjusted for the effective field in the sample through $\tilde{E}_{\text{THz}} = \beta E_{\text{THz}}$, where β quantifies the screening of the vacuum THz field by the sample. The solution of these coupled differential equations yields the dynamics of each individual mode for a given field, and the total structural dynamics is given by the superposition of the two eigenvectors.

To explore the specific dynamics for our STO film, we utilize first principle calculations to quantify the sizes of the anharmonic coupling terms and the mode effective charges, employing a previously established approach based on density functional theory^{41,42}. We used ellipsometry measurements of our sample (Fig. 3a) to obtain the frequency and lifetime of the soft mode and combined literature and ellipsometry measurements for the TO₂ mode frequency^{39,43}. We relied on our trXRD data to model the TO₂ linewidth γ_2 and the overall field scaling β . Because the DFT

calculations are performed at 0 K, we tuned several of the more sensitive parameters from equations (1,2) ($\kappa_1, \psi_{12}, \chi$) to best match our experimental results at 100 K. We extract the unit cell structure and eigenvectors from our first principle calculations and use these results to transform the calculated phonon amplitudes into the expected x-ray scattering change of the (2 -2 3) reflection. Numerical details are given in the Methods section and supplement.

The results of our model in comparison with the experimental data are shown in Fig. 4. We readily reproduce the primary time-domain features (black lines in Fig. 4a), including the overall waveform shape as well as the peak saturation (Fig. 4b) and appearance of 5.15 THz mode (Fig. 4d). Figure 4e shows the resultant quartic potential for the soft mode Q_1 (blue, solid) when Q_2 is zero. The gray shaded region indicates the maximum excursion in our simulation when using the highest experimental THz field strength (880 ± 50 kV/cm), reaching about $Q_1 = -0.2 \text{ \AA} \sqrt{AMU}$. This corresponds to 1.3 pm real space motion parallel to the THz driving field for the Ti ions (see supplement for full atomic displacement information). For reference, in the prototypical ferroelectric perovskite BaTiO₃, the Ti ion moves 9 pm from the cubic site at 300 K⁴⁴.

We can gain more physical insight into the dynamics of the system by further simplifying our model. The oscillator strength for the soft mode is much larger than the TO₂ mode, hence $Q_1 \gg Q_2$ while the THz driving field is present. Moreover, energy transfer into this mode is favored because Q_1 is in resonance with the THz field. Therefore, the most dominant nonlinear driving terms in the equations of motion must be $-\kappa_1 Q_1^3$ and $-\psi_{12} Q_1^3$ for Q_1 and Q_2 , respectively. This allows us to simplify the equations of motion (2) as

$$\ddot{Q}_1 + \gamma_1 \dot{Q}_1 + \tilde{\omega}_1^2 Q_1 = Z_1^* \tilde{E}_{\text{THz}}, \quad (3a)$$

$$(3b)$$

$$\ddot{Q}_2 + \gamma_2 \dot{Q}_2 + \omega_2^2 Q_2 = Z_2^* \tilde{E}_{\text{THz}} - \psi_{12} Q_1^3,$$

where $\tilde{\omega}^2 = (\omega_1^2 + \kappa_1 Q_1^2)$ describes an amplitude-dependent frequency. Equation (3a) thus describes the motion of a driven oscillator with a frequency that increases with mode amplitude. This is consistent with the soft mode hardening effect observed in STO in static electric fields⁴⁵. We attribute the saturation and steepening of the peak trXRD signal to this effect: for large amplitudes, the resonance condition for Q_1 shifts above the bandwidth of the THz driving field, thereby reducing energy transfer from the THz field into the soft mode as Q_1 increases.

For the equation of motion of Q_2 , equation (3b), we neglect any nonlinear terms. Equation (3b) then corresponds to a harmonic oscillator with two driving terms: the THz electric field $\tilde{E}_{\text{THz}}$ and the other phonon mode Q_1 . The latter term contains the third power of Q_1 which results in the generation of additional spectral components at the third harmonic of the spectrum of Q_1 . This extends the fundamental bandwidth of the driving THz field to overlap with the Q_2 mode and results in more efficient energy transfer. This simple picture is in excellent agreement with our observation of the 5.15 THz TO_2 resonance driven by the anharmonic soft mode. We find that we best reproduce our data with $\kappa_1 = 8900 \text{ THz}^2 \text{\AA}^{-2} \text{AMU}^{-1}$, $\psi_{12} = -4000 \text{ THz}^2 \text{\AA}^{-2} \text{AMU}^{-1}$, and $\chi = 4000 \text{ THz}^2 \text{\AA}^{-2} \text{AMU}^{-1}$. For reference, Katayama et al. use a single quartic potential to model the soft mode in STO films, yielding the equivalent of $\kappa_1 = 19000 \text{ THz}^2 \text{\AA}^{-2} \text{AMU}^{-1}$ in reasonable agreement with our results⁴⁶. With only two nonlinear terms therefore we can explain the field-dependent observations in our x-ray scattering signal.

It is striking that we observe a response at the 7.6 THz mode because in bulk STO this phonon is silent (neither IR nor Raman active)³⁸. Despite the tetragonal structure of our film, ellipsometry measurements at room temperature show no IR activity in this frequency window

(see supplement) nor do measurements on a different STO/LSAT system at 5 K and 300 K³⁹. Our model provides a ready channel to transfer energy from one phonon mode to another via anharmonic coupling and could be expanded to incorporate this additional mode. Thus, our novel excitation method has the potential to stimulate otherwise inaccessible phonon modes via optical techniques.

In conclusion, we have reported direct structural evidence of phonon up-conversion in the perovskite incipient ferroelectric STO. Intense single-cycle THz radiation couples directly to the soft mode via infrared absorption and drives the system far from equilibrium into the strongly nonlinear regime. Subsequent energy coupling into high frequency phonon modes at 5.15 and 7.6 THz occurs, as directly observed in time-resolved x-ray diffraction measurements. We can capture the nonlinear response by invoking terms up to fourth-order coupling between the soft mode and the 5.15 THz mode, reproducing with high fidelity the observed x-ray scattering signal. All-optical measurements have recently reported similar effects in Bi₂Se₃⁴⁷ and CdWO₄⁴⁸ but the presence of free carriers in the former system complicates the driving mechanism; in contrast, STO has a large bandgap and no free carriers, enabling unambiguous attribution of the excitation to nonlinear phononics. Moreover, our x-ray measurements on STO provide a direct structural probe compared to optical techniques.

Our results demonstrate a mechanism to transfer energy into higher energy modes by exploiting the nonlinearities that couple phonons. It has the potential even to channel energy into so-called “silent modes” that are neither IR- nor Raman-active, as we here report energy coupling into a mode that is in bulk, silent. Moreover, current tools for enhancing and engineering THz interactions with matter^{32,49} may now be leveraged to control higher-frequency,

mid-infrared modes via this nonlinear phonon up-conversion process. Beyond the scope of nonlinear phononics, it is furthermore remarkable that we can drive the soft mode phonon so far into the nonlinear regime, establishing the potential to drive soft mode-mediated domain switching in related ferroelectric perovskites⁵⁰.

References

1. Cavalleri, A. *et al.* Femtosecond Structural Dynamics in VO₂ during an Ultrafast Solid-Solid Phase Transition. *Phys. Rev. Lett.* **87**, 237401 (2001).
2. Kimel, A. V., Kirilyuk, A., Tsvetkov, A., Pisarev, R. V. & Rasing, T. Laser-induced ultrafast spin reorientation in the antiferromagnet TmFeO₃. *Nature* **429**, 850–853 (2004).
3. Beaud, P. *et al.* A time-dependent order parameter for ultrafast photoinduced phase transitions. *Nat. Mater.* **13**, 923–7 (2014).
4. Chuang, Y. D. *et al.* Real-Time Manifestation of Strongly Coupled Spin and Charge Order Parameters in Stripe-Ordered La_{1.75}Sr_{0.25}NiO₄ Nickelate Crystals Using Time-Resolved Resonant X-Ray Diffraction. *Phys. Rev. Lett.* **110**, 127404 (2013).
5. Schlom, D. G. *et al.* Elastic strain engineering of ferroic oxides. *MRS Bull.* **39**, 118–130 (2014).
6. Tomioka, Y., Asamitsu, A., Kuwahara, H., Moritomo, Y. & Tokura, Y. Magnetic-field-induced metal-insulator phenomena in Pr_{1-x}Ca_xMnO₃ with controlled charge-ordering instability. *Phys. Rev. B* **53**, R1689–R1692 (1996).
7. Haeni, J. H. *et al.* Room-temperature ferroelectricity in strained SrTiO₃. *Nature* **430**, 758–761 (2004).
8. Chen, X. J. *et al.* Enhancement of superconductivity by pressure-driven competition in electronic order. *Nature* **466**, 950–953 (2010).
9. Rini, M. *et al.* Control of the electronic phase of a manganite by mode-selective vibrational excitation. *Nature* **449**, 72–74 (2007).

10. Först, M., Mankowsky, R. & Cavalleri, A. Mode-selective control of the crystal lattice. *Acc. Chem. Res.* **48**, 380–387 (2015).
11. Först, M. *et al.* Nonlinear phononics: A new ultrafast route to lattice control. *Nat. Phys.* **7**, 854–856 (2011).
12. Juraschek, D. M. & Maehrlein, S. F. Sum-frequency ionic Raman scattering. *Phys. Rev. B* **97**, 174302 (2018).
13. Caviglia, A. D. *et al.* Ultrafast strain engineering in complex oxide heterostructures. *Phys. Rev. Lett.* **108**, (2012).
14. Lee, J. J. *et al.* Interfacial mode coupling as the origin of the enhancement of T_{c} in FeSe films on SrTiO₃. *Nature* **515**, 245–248 (2014).
15. Gor'kov, L. P. Phonon mechanism in the most dilute superconductor n -type SrTiO₃. *Proc. Natl. Acad. Sci.* **113**, 4646–4651 (2016).
16. Rowley, S. E. *et al.* Ferroelectric quantum criticality. *Nat. Phys.* **10**, 367–372 (2014).
17. Edge, J. M., Kedem, Y., Aschauer, U., Spaldin, N. A. & Balatsky, A. V. Quantum Critical Origin of the Superconducting Dome in SrTiO₃. *Phys. Rev. Lett.* **115**, 247002 (2015).
18. Barker, A. S. & Tinkham, M. Far-Infrared Ferroelectric Vibration Mode in SrTiO₃. *Phys. Rev.* **125**, 1527–1530 (1962).
19. Cowley, R. A. Temperature dependence of a transverse optic mode in strontium titanate. *Phys. Rev. Lett.* **9**, 159–161 (1962).
20. Yamada, Y. & Shirane, G. Neutron scattering and nature of the soft optical phonon in SrTiO₃. *Journal of the Physical Society of Japan* **26**, 396–403 (1969).
21. Cochran, W. Crystal stability and the theory of ferroelectricity. *Phys. Rev. Lett.* **3**, 412–414

- (1959).
22. Cochran, W. Crystal stability and the theory of ferroelectricity. *Adv. Phys.* **9**, 387–423 (1960).
 23. Lines, M. E. & Glass, A. M. *Principles and Applications of Ferroelectrics and Related Materials*. (Oxford University Press, 2001).
doi:10.1093/acprof:oso/9780198507789.001.0001
 24. Müller, K. A. & Burkard, H. SrTiO₃: An intrinsic quantum paraelectric below 4 K. *Phys. Rev. B* **19**, 3593–3602 (1979).
 25. De Silvestri, S. *et al.* Femtosecond time-resolved measurements of optic phonon dephasing by impulsive stimulated raman scattering in α -perylene crystal from 20 to 300 K. *Chem. Phys. Lett.* **116**, 146–152 (1985).
 26. Cho, G. C., Kütt, W. & Kurz, H. Subpicosecond time-resolved coherent-phonon oscillations in GaAs. *Phys. Rev. Lett.* **65**, 764–766 (1990).
 27. Kohmoto, T., Tada, K., Moriyasu, T. & Fukuda, Y. Observation of coherent phonons in strontium titanate: Structural phase transition and ultrafast dynamics of the soft modes. *Phys. Rev. B* **74**, 064303 (2006).
 28. Kubacka, T. *et al.* Large-Amplitude Spin Dynamics Driven by a THz Pulse in Resonance with an Electromagnon. *Science* **343**, 1333–1336 (2014).
 29. Mankowsky, R. *et al.* Nonlinear lattice dynamics as a basis for enhanced superconductivity in YBa₂Cu₃O_{6.5}. *Nature* **516**, 71–73 (2014).
 30. Grübel, S. *et al.* Ultrafast x-ray diffraction of a ferroelectric soft mode driven by broadband terahertz pulses. *arXiv* 1602.05435 (2016).

31. Marsik, P. *et al.* Terahertz ellipsometry study of the soft mode behavior in ultrathin SrTiO₃ films. *Appl. Phys. Lett.* **108**, 052901 (2016).
32. Kozina, M. *et al.* Local Terahertz Field Enhancement for Time-Resolved X-ray Diffraction. *Appl. Phys. Lett.* **110**, 081106 (2017).
33. Spitzer, W. G., Miller, R. C., Kleinman, D. A. & Howarth, L. E. Far infrared dielectric dispersion in BaTiO₃, SrTiO₃, and TiO₂. *Phys. Rev.* **126**, 1710–1721 (1962).
34. Barker, A. S. Temperature dependence of the transverse and longitudinal optic mode frequencies and charges in SrTiO₃ and BaTiO₃. *Phys. Rev.* **145**, 391–399 (1966).
35. Kozina, M. *et al.* Ultrafast X-ray diffraction probe of terahertz field-driven soft mode dynamics in SrTiO₃. *Struct. Dyn.* **4**, 054301 (2017).
36. Cowley, R. A. Lattice Dynamics and Phase Transitions of Strontium Titanate. *Phys. Rev.* **134**, A981–A997 (1964).
37. Servoin, J. L., Luspin, Y. & Gervais, F. Infrared dispersion in SrTiO₃ at high temperature. *Phys. Rev. B* **22**, 5501–5506 (1980).
38. Denisov, V. N., Mavrin, B. N., Podobedov, V. B. & Scott, J. F. Hyper-Raman Spectra and Frequency Dependence of Soft Mode Damping in SrTiO₃. *J. Raman Spectrosc.* **14**, 276–283 (1983).
39. Nuzhnyy, D. *et al.* Polar phonons in some compressively stressed epitaxial and polycrystalline SrTiO₃ thin films. *J. Electroceramics* **22**, 297–301 (2009).
40. Gonze, X. & Lee, C. Dynamical matrices, Born effective charges, dielectric permittivity tensors, and interatomic force constants from density-functional perturbation theory. *Phys. Rev. B* **55**, 10355–10367 (1997).

41. von Hoegen, A., Mankowsky, R., Fechner, M., Först, M. & Cavalleri, A. Probing the interatomic potential of solids with strong-field nonlinear phononics. *Nature* **555**, 79–82 (2018).
42. Cartella, A., Nova, T. F., Fechner, M., Merlin, R. & Cavalleri, A. Parametric amplification of optical phonons. *arXiv* 1708.09231 (2017).
43. Petzelt, J. *et al.* Dielectric, infrared, and Raman response of undoped SrTiO₃ ceramics: Evidence of polar grain boundaries. *Phys. Rev. B* **64**, 184111 (2001).
44. Kwei, G. H., Lawson, a C., Billinge, S. J. L. & Cheong, S. W. Structures of the ferroelectric phases of barium titanate. *J. Phys. Chem.* **97**, 2368–2377 (1993).
45. Akimov, I. A., Sirenko, A. A., Clark, A. M., Hao, J.-H. & Xi, X. X. Electric-Field-Induced Soft-Mode Hardening in SrTiO₃ Films. *Phys. Rev. Lett.* **84**, 4625–4628 (2000).
46. Katayama, I. *et al.* Ferroelectric soft mode in a SrTiO₃ thin film impulsively driven to the anharmonic regime using intense picosecond terahertz pulses. *Phys. Rev. Lett.* **108**, 097401 (2012).
47. Melnikov, A. A. *et al.* Coherent phonons in a Bi₂Se₃ film generated by an intense single-cycle THz pulse. *Phys. Rev. B* **97**, 214304 (2018).
48. Johnson, J. A., Ellsworth, N. R. & Salmans, P. D. High-Field THz Lattice Control Via Anharmonic Vibrational Coupling. *Int. Conf. Ultrafast Phenom.* OSA Technical Digest, UW4A.11 (2016).
49. Gray, A. X. *et al.* Ultrafast terahertz field control of electronic and structural interactions in vanadium dioxide. *Phys. Rev. B* **98**, 045104 (2018).
50. Qi, T., Shin, Y.-H., Yeh, K.-L., Nelson, K. A. & Rappe, A. M. Collective Coherent Control:

- Synchronization of Polarization in Ferroelectric PbTiO₃ by Shaped THz Fields. *Phys. Rev. Lett.* **102**, 247603 (2009).
51. Chollet, M. *et al.* The X-ray Pump–Probe instrument at the Linac Coherent Light Source. *J. Synchrotron Radiat.* **22**, 503–507 (2015).
 52. Zhu, D. *et al.* Performance of a beam-multiplexing diamond crystal monochromator at the Linac Coherent Light Source. *Rev. Sci. Instrum.* **85**, 063106 (2014).
 53. Hebling, J., Almasi, G., Kozma, I. & Kuhl, J. Velocity matching by pulse front tilting for large area THz-pulse generation. *Opt. Express* **10**, 1161–1166 (2002).
 54. Yeh, K. L., Hoffmann, M. C., Hebling, J. & Nelson, K. A. Generation of 10 μ J ultrashort terahertz pulses by optical rectification. *Appl. Phys. Lett.* **90**, 171121 (2007).
 55. Minitti, M. P. *et al.* Optical laser systems at the Linac Coherent Light Source. *J. Synchrotron Radiat.* **22**, 526–531 (2015).
 56. Bionta, M. R. *et al.* Spectral encoding of x-ray/optical relative delay. *Opt. Express* **19**, 21855 (2011).
 57. Herrmann, S. *et al.* CSPAD-140k: A versatile detector for LCLS experiments. *Nucl. Instruments Methods Phys. Res. Sect. A* **718**, 550–553 (2013).
 58. Blaj, G. *et al.* X-ray detectors at the Linac Coherent Light Source. *J. Synchrotron Radiat.* **22**, 577–583 (2015).
 59. Sirenko, A. A. *et al.* Soft-mode hardening in SrTiO₃ thin films. *Nature* **404**, 373–376 (2000).
 60. Henke, B. L., Gullikson, E. M. & Davis, J. C. X-Ray interactions: photoabsorption, scattering, transmission, and reflection at $E = 50\text{--}30,000$ eV, $Z = 1\text{--}92$. *Atomic Data and*

Nuclear Data Tables **54**, 181–342 (1993).

61. Brown, P. J., Fox, A. G., Maslen, E. N., O’Keefe, M. A. & Willis, B. T. M. *Intensity of diffracted intensities. International Tables for Crystallography C*, (International Union of Crystallography, 2006).
62. Berkeley Lab. The Atomic Scattering Factor Files. Available at:
http://henke.lbl.gov/optical_constants/asf.html. (Accessed: 25th July 2018)
63. Giannozzi, P. *et al.* QUANTUM ESPRESSO: a modular and open-source software project for quantum simulations of materials. *J. Phys. Condens. Matter* **21**, 395502 (2009).
64. Monkhorst, H. J. & Pack, J. D. Special points for Brillouin-zone integrations. *Phys. Rev. B* **13**, 5188–5192 (1976).
65. Perdew, J. P. *et al.* Restoring the Density-Gradient Expansion for Exchange in Solids and Surfaces. *Phys. Rev. Lett.* **100**, 136406 (2008).
66. Baroni, S., de Gironcoli, S., Dal Corso, A. & Giannozzi, P. Phonons and related crystal properties from density-functional perturbation theory. *Rev. Mod. Phys.* **73**, 515–562 (2001).
67. Resta, R. Macroscopic polarization in crystalline dielectrics: the geometric phase approach. *Rev. Mod. Phys.* **66**, 899–915 (1994).

Acknowledgements

Use of the Linac Coherent Light Source (LCLS), SLAC National Accelerator Laboratory, is supported by the U.S. Department of Energy, Office of Science, Office of Basic Energy Sciences, under Contract No. DE-AC02-76SF00515. M.K. and M.C.H. are supported by the U.S. Department of Energy, Office of Science, Office of Basic Energy Sciences, under Award No. 2015-SLAC-100238-Funding. U.S. acknowledges support from the National Center of Competence in Research: Ultrafast Science and Technology (NCCR MUST) of the Swiss National Science Foundation. S.B. acknowledges support from the Knut and Alice Wallenberg Foundation. Work at the University of Fribourg was supported by the Schweizer Nationalfonds (SNF) by Grant No. 200020-172611. M.K. and M.C.H. extend thanks to William Chueh and Antonio Bacig for annealing the sample and to Ziran Wu for assistance with the THz experiments. M.F. extends thanks to Michael Först for fruitful discussions about modelling the STO system.

Author Contributions

M.K. and M.C.H. conceived the experiment and performed the final data analysis. M.F. provided DFT calculations and theory support. T.v.D. and S.B. helped with on-line data analysis. M.K., M.C.H., J.M.G. and D.Z. performed the time-resolved x-ray experiment. U.S. provided sample expertise and additional x-ray data. M.R. prepared the sample. P.M. and C.B. made the THz ellipsometry measurements of the sample. The paper was written by M.K. and M.C.H., with substantial contributions from M.F., U.S., and S.B., as well as with discussions from other authors.

Competing Interests

The authors declare no competing interests.

Methods

Experimental Details. The measurements were carried out at the XPP instrument of the Linac Coherent Light Source⁵¹ in monochromatic mode. The x-rays were tuned to 8 keV with a bandwidth of ~1 eV after the monochromator⁵² and were 30 fs full-width, half-max (FWHM) in duration at 120 Hz repetition rate.

Single-cycle THz radiation was generated through optical rectification of 800 nm femtosecond pulses in LiNbO₃ using the tilted pulse-front technique^{53,54}. We used up to 20 mJ pulse energy at 120 Hz repetition rate and 100 fs FWHM pulse duration⁵⁵. The THz pulses were focused into the sample using a three parabolic mirror geometry with intermediate focus. A pair of wiregrid polarizers was used to controllably attenuate the THz field strength without affecting the pulse shape by rotating the first polarizer. We characterized the electric field component of the THz pulse at the sample location using electro-optic sampling in 50 μm thick (110) cut GaP crystals and found the peak electric field to be 880 ± 50 kV/cm. The peak frequency of the THz pump pulse is at 0.75 THz with components extending to 2.5 THz.

The intrinsic timing jitter between the THz pump and x-ray probe pulses was mitigated using a spectral encoding technique⁵⁶, and data were temporally binned to 25 fs resolution. We collected x-ray scattering data with the CSPAD area detector^{57,58} and normalized with a shot-by-shot x-ray intensity monitor as well as by shots when the pump laser was not present. A typical dataset included 60k (36k) x-ray shots for the laser on (off) after excluding shots with low x-ray intensity or poor spectral encoding signal. We fixed the detector and sample in one scattering geometry and integrated over a region of interest on the detector to estimate the diffraction peak intensity. Error bars on the trXRD signal were estimated from the standard deviation of the signal for each time delay bin.

We maintained the sample at ~ 100 K using a nitrogen cryostream (Oxford Instruments Cryojet 5). The sample temperature thus has a lower bound of 100 K but could be at most 10 K higher because the temperature is measured at the cryostream jet output.

Sample Details. Our sample consisted of a 50 nm thick STO film on an $(\text{LaAlO}_3)_{0.3}(\text{Sr}_2\text{TaAlO}_6)_{0.7}$ (LSAT) substrate. It was annealed at 1200 C for 12 h to diminish the effects of the substrate on the phonon dispersion^{31,59}. X-ray diffraction measurements before annealing indicated the film was compressed from bulk STO in-plane to match the substrate and expanded in the cross-plane direction. The lattice parameters were $a_0=3.867\text{\AA}$ and $c_0=3.925\text{\AA}$. See supplement for static x-ray diffraction data.

Structure Factor Calculations. Experimentally we measure the diffraction peak intensity I for a given set of lattice planes corresponding to the reciprocal lattice vector $\mathbf{G}$. As the ions in the lattice respond to the THz excitation, the x-ray scattering intensity changes. The intensity $I \propto |F|^2$, where the structure factor $F = \sum_j f_j e^{-i\mathbf{r}_j \cdot \mathbf{G}}$. Here f_j and $\mathbf{r}_j$ are the atomic scattering factor and position vector for the ion j , respectively; the sum runs over all ions in the unit cell. Note that the f_j are complex and extrapolated from tabulated values^{60–62} for the experimental x-ray energy of 8 keV. In our model we calculate the phonon amplitudes Q_i , which we use to extract the change in ion position through the mode eigenvectors ξ_{ijk} via $r_{jk} = r_{jk_0} + \sum_i \xi_{ijk} Q_i$, where i, j, k index respectively the phonon branch, ion, and coordinate direction, and r_{jk_0} is the equilibrium position of the ion. Denoting F_0 as the structure factor in equilibrium, the fractional change in x-ray scattering intensity $\Delta I/I = |F|^2/|F_0|^2 - 1$. We calculate the STO unit cell structure r_{jk_0} and eigenvectors ξ_{ijk} using density functional theory as described below in the Numerical Details and combine this information with the phonon amplitudes Q_i to compute the expected relative change in x-ray scattering intensity.

Numerical Details. Our calculations were carried out using density functional theory (DFT) as implemented in the Quantum Espresso code⁶³. We used projected augmented wave (PAW) pseudopotentials, which contain as valence states the $4s^2 5s^2 4p^6$ for Sr, $3s^2 3p^6 3d^2 4s^2$ for Titanium and $2s^2 2p^4$ for Oxygen. In all computations, we sampled the Brillouin zone by a $14 \times 14 \times 14$ k-point mesh generated with the Monkhorst and Pack scheme⁶⁴ and placed a cutoff energy on the wavefunction of 45 Rydbergs. As an approximation for the exchange correlation functional, we applied PBEsol⁶⁵. All total energy computations have been reiterated until the change in energy became less than 10^{-10} Rydberg. As phonon computations require a force-free groundstate, we performed structural relaxation until the forces acting on individual ions became lower than $5 \mu\text{Ry}/a_0$. The phonon frequencies and eigenvectors have been obtained by density functional perturbation theory⁶⁶ calculations, whereas the mode effective charges are computed utilizing

the modern theory of polarization^{40,67}. Finally, we compute the anharmonic coupling coefficients from frozen phonon calculations. Thereby, we modulate the structure with an appropriate superposition of the phonon eigenvectors and compute the resulting total energies. The anharmonic coefficients listed in equation (1) are then obtained by a least mean squares fit of the multidimensional potential landscape. We note that in our fitting procedure we also take into account higher order terms than those given in equation (1), which due to their size, however, for the specific case of STO can be neglected. In order to fit our experimental data, we tweaked several of the anharmonic coefficients as noted in the main text, recognizing that the DFT calculations are performed at 0 K while our experimental measurements were collected ~100K. Numerical values for the STO unit cell structure, eigenvectors, and model coefficients from equations (1-3) are listed in the supplement.

Data Availability. The data that support the findings of this study are available from the corresponding author upon reasonable request.