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Characterization of iLGADs using soft X-rays

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ABSTRACT: Experiments at synchrotron radiation sources and X-ray Free-Electron Lasers in the soft X-ray energy range (250 eV–2 keV) stand to benefit from the adaptation of the hybrid silicon detector technology for low energy photons. Inverse Low Gain Avalanche Diode (iLGAD) sensors provide an internal gain, enhancing the signal-to-noise ratio and allowing single photon detection below 1 keV using hybrid detectors. In addition, an optimization of the entrance window of these sensors enhances their quantum efficiency (QE).

In this work, the QE and the gain of a batch of different iLGAD diodes with optimized entrance windows were characterized using soft X-rays at the Surface/Interface:Microscopy beamline of the Swiss Light Source synchrotron. Above 250 eV, the QE is larger than 55% for all sensor variations, while the charge collection efficiency is close to 100%. The average gain depends on the gain layer design of the iLGADs and increases with photon energy. A fitting procedure is introduced to extract the multiplication factor as a function of the absorption depth of X-ray photons inside the sensors. In particular, the multiplication factors for electron- and hole-triggered avalanches are estimated, corresponding to photon absorption beyond or before the gain layer, respectively.

Keywords: Charge transport and multiplication in solid media; Hybrid detectors; Solid state detectors; X-ray detectors

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1 Introduction

Hybrid detectors are widely used for the detection of hard and tender X-rays at synchrotron radiation sources and X-ray Free-Electron Lasers (XFELs) thanks to their versatility and robustness, combined with high dynamic range, fast frame rate, and the possibility of building large detector systems by tiling individual modules [1]. Microscopy and spectroscopic techniques applied to the soft X-ray range (250 eV–2 keV) would also benefit from such characteristics to study material properties by exploiting the K absorption edges of light organic elements (e.g. C, N, O) or the L-edges of 3d transition metals (e.g. Cu, Fe, Ti).

However, to make hybrid detectors suitable for applications in the soft X-ray energy range, the following challenges must be overcome:

- The low quantum efficiency (QE) of planar silicon sensors below 2 keV, due to the sub- μm attenuation length of soft X-rays in solids (figure 1(a)). Typical p-in-n silicon sensors for photon science (figure 1(b)) feature an X-ray entrance window composed of a ~ 1–2 μm aluminum layer and ~ 1–2 μm of highly doped n-type region. Soft X-rays can be easily absorbed in the insensitive aluminum layer. In addition, the charge carriers generated in the undepleted part of the n-type region may recombine before they generate a signal in the readout electrode.
- The low signal-to-noise ratio (SNR) due to the small number of charge carriers created by soft X-rays. For example, 250 eV photons produce ~ 70 electron-hole pairs, which is comparable with the electronic noise of typical hybrid detectors. Considering the noise of state-of-the-art

hybrid detectors ($\sim 30\text{--}100\,\text{e}^-$ equivalent noise charge RMS [3–5]), the minimal detectable energy of single X-ray photons with an SNR greater than 5 is $\sim 600-1800\,\text{eV}$.

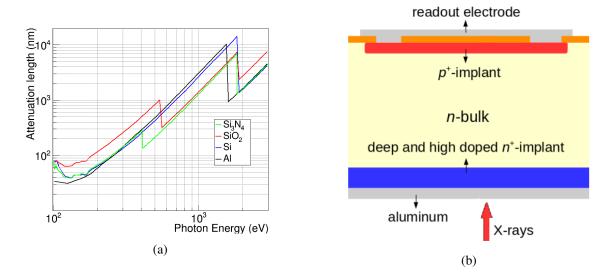


Figure 1. (a) Attenuation length as a function of the photon energy in Al, Si_3N_4 , SiO_2 , and Si between 100 eV and 3 keV [2] and (b) cross section of a conventional planar silicon sensor.

The Paul Scherrer Institut (PSI), in collaboration with Fondazione Bruno Kessler (FBK), is developing planar silicon sensors with optimized entrance windows (EW) for enhanced QE and, in parallel, inverse Low-Gain Avalanche Diode (iLGAD) sensors, to achieve single-photon detection in the soft X-rays range. The optimized EW has now been implemented on the iLGAD sensors.

iLGADs [6] are manufactured on p-type silicon wafers, with an n^+ contact and a p gain layer on the backplane, where the EW is located, and readout channels formed segmenting a p^+ region on the front side (figure 2). When an external reverse bias is applied, the electrons drift towards the backplane, while the holes are collected by the readout electrodes at the opposite side of the sensor. The p-type gain layer is formed by boron implantation confined to within 1 μ m of the surface of the Si bulk. In this region, an electric field greater than 300 kV/cm accelerates charge carriers so that they produce further carriers via impact ionization. The increase in the signal amplitude improves the SNR if the detector noise is not dominated by the shot noise in the sensor. In iLGADs, the gain layer uniformly covers the entire sensor area. As a result, the fraction of the sensor area of a readout channel where the charge multiplication occurs is 100%, regardless of the segmentation of the readout electrodes. This is an advantage compared to conventional LGADs, which have a no-gain region between segmented readout electrodes that is at least 30 µm wide [7], though this can be reduced in the range $2-5 \mu m$ using trench-isolation [8, 9]. The spatial uniformity of the multiplication is essential for X-ray position-sensitive detectors, in particular for experiments that require high spatial resolution by interpolation using charge sharing [10]. The iLGAD sensors used in this work coupled to the MÖNCH detector demonstrated single-photon detection down to a photon energy of 452 eV, when illuminated with fluorescence X-rays [11].

To optimize the EW, the metalization is replaced with a thin passivation, composed of Si_3N_4 and SiO_2 layers (figure 2) [11]. In particular, the SiO_2 layer is used to reduce the recombination effects of

charge carriers at the surface by compensating the silicon dangling bonds with the thermally grown oxide. The n^+ implant at the backplane is made as shallow as possible in order to reduce the width of the undepleted low-field region where charge carriers generated by X-rays can recombine. In this way, the charge carriers (holes in the iLGAD sensor) are able to diffuse out of the undepleted region and drift to the collecting electrode on the opposite side under the influence of a strong electric field. An investigation of the QE of planar silicon sensors with optimized EW using 405 nm UV light has been reported in [12].

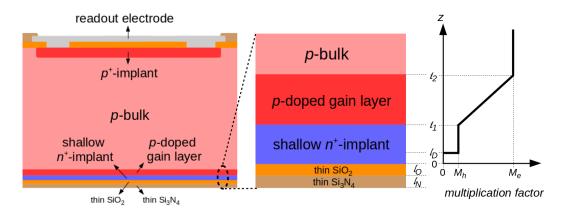


Figure 2. Cross section of an iLGAD sensor with optimized EW.

This paper presents measurements of the QE and the average gain of different variations of iLGADs combined with improved EW. The measurements span the photon energy range between 200 eV and 1 keV. Models describing the QE and average gain as functions of the energy are discussed and phenomenological models are also introduced for the charge multiplication factor as a function of the depth where the charge carriers are generated. The extracted depth-dependent multiplication factor is of fundamental importance to understand the spectral response of hybrid pixel detectors using iLGAD sensors to monoenergetic X-ray photons. It will be employed in device simulations to reproduce the measured spectral response of charge-integrating detectors, which will be discussed in a separate paper. The investigation of the multiplication factor as a function of the photon absorption depth in the sensors under test and the study of their spectral response are relevant steps for the optimization of the design of iLGADs for soft X-ray detection using the hybrid detector technology.

2 Materials and methods

2.1 Sensors under test

In order to characterize the quantum efficiency and gain of several sensor variations, pairs of X-ray sensing photodiodes, consisting of an iLGAD diode and an n^+ -p- p^+ diode without gain layer manufactured by FBK, were tested. Both sensors were fabricated on the same silicon wafer and underwent the same production process, except for the implantation of the gain layer. The diodes are 275 μ m thick and their active area is 4 mm², surrounded by a current-collection ring (CCR) and ten floating guard rings to prevent breakdown at low bias voltages. Each sensor is glued onto a small PCB and wire-bonded to read out the photocurrent during the measurement (figure 3).

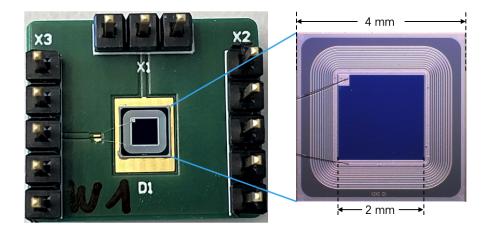


Figure 3. Top view of the n^+ -p- p^+ and iLGAD diodes under test. The active area is 4 mm^2 and is surrounded by one current-collection ring and ten floating guard rings.

Wafer identifier	n ⁺ implant	p gain layer design	p gain layer dose	
W15	Standard	Standard	Low	
W17	Shallow	Standard	Medium	
W5	Shallow	Shallow	Low	
W9	Shallow	Shallow	High	
W19	Ultra-shallow	Ultra-shallow	Medium	
W13	Ultra-shallow	Ultra-shallow	High	

Table 1. Characteristics of the different diodes under test as provided by the manufacturer.

Among the process splits, two n^+ -implant designs (shallow and ultra-shallow) at the EW and three gain layer profiles with different depths (namely standard, shallow, and ultra-shallow) and using different implantation doses have been investigated. Table 1 shows a summary of the n^+ and gain layer designs of the wafers under study.

2.2 Experimental setup

The measurements were carried out at the Surface/Interface:Microscopy (SIM) beamline of the Swiss Light Source (SLS) synchrotron. The beamline covers an energy range between 90 eV and 2 keV, with an X-ray energy resolution $E/\Delta E > 5000$ [13]. The experimental setup is shown in figure 4.

The diodes under test were mounted on a motorized stage inside a vacuum chamber at the FLASH endstation of the beamline. Three diodes fabricated by FBK were mounted at a time on the stage, along with a calibrated photodiode for which the quantum efficiency $QE_0(E)$ is known in the energy range of interest. The temperature of the diodes was stabilized at 20° C with a liquid cooling system. The diodes were reverse biased at 300 V simultaneously with a Keithley 6517B power supply and fully depleted.

The X-ray beam enters the vacuum chamber through a few mm wide pinhole. It is collimated and then shaped by an Order Sorting Aperture (OSA), a $65 \,\mu m$ diameter pinhole positioned in front of the diode under test. The stage could be moved to center the beam on the active area of each

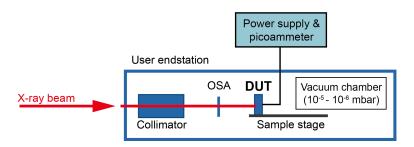


Figure 4. Scheme of the experimental setup at the endstation of the SIM beamline of the SLS. The beamline structure is shown in [14].

sensor. After proper alignment, the beam spot was fully contained in the active areas of the diodes. The OSA could be moved laterally to block the beam and measure the dark current.

At each photon energy E, the currents with X-ray illumination I^{meas} and the dark currents I^{dark} were measured for the three diodes under test and the calibrated photodiode. Four Keithley picoammeters were employed for the measurements. The photocurrent is calculated as $I^{\gamma} = I^{\text{meas}} - I^{\text{dark}}$ and is denoted as I^{γ}_{pin} , $I^{\gamma}_{\text{iLGAD}}$, and I^{γ}_{0} for the n^{+} -p-p+ diodes, the iLGAD diodes, and calibrated photodiode, respectively. The photocurrent of a diode depends on the photon energy E and is proportional to the photon rate Φ of the X-ray beam. The fluctuations of Φ at a given photon energy during the experiments can be negletcted since the current of the electron beam in the synchrotron ring fluctuates by less than 0.25%.

2.3 Data analysis methods

2.3.1 Quantum efficiency

For the measurements of the QE of the iLGADs, we rely on the measurement of the n^+ -p- p^+ diodes photocurrents, assuming that the QE is identical for devices fabricated on the same wafer with the same passivation and profile of the n^+ implant. The QE of an X-ray sensing diode (without charge multiplication) is determined by the ratio between the photocurrent and the current generated if all photons were absorbed inside the active region of the sensor and all carriers produced were collected:

$$QE = \frac{I^{\gamma}(E, \Phi)}{q_0 \frac{\Phi E}{3.6 \text{ eV}}},$$
(2.1)

where q_0 is the elementary charge, E is the photon energy and Φ is the photon flux. Since this applies to both the n^+ -p- p^+ diodes and the calibrated diode, the QE for each n^+ -p- p^+ diode can be calculated by comparing I_{pin}^{γ} and I_0^{γ} at the same energy and flux and normalizing with the QE of the calibrated diode QE₀:

$$QE_{pin} = \frac{I_{pin}^{\gamma}(E, \Phi)}{I_{0}^{\gamma}(E, \Phi)} QE_{0}$$
(2.2)

For soft X-rays at normal incidence, the QE of the diode with SiO₂ and Si₃N₄ layers at the EW as a function of the photon energy can be expressed as:

$$QE(E) = \exp\left(-\frac{l_{N}}{\lambda_{N}(E)}\right) \exp\left(-\frac{l_{O}}{\lambda_{O}(E)}\right) \int_{0}^{L} CCE(z) \cdot \frac{1}{\lambda_{Si}(E)} \exp\left(-\frac{z}{\lambda_{Si}(E)}\right) dz, \quad (2.3)$$

where $l_{\rm N}$ and $l_{\rm O}$ are the thicknesses of the Si₃N₄ and SiO₂ layers; $\lambda_{\rm N}$, $\lambda_{\rm O}$, $\lambda_{\rm Si}$ are the attenuation lengths of X-ray photons in Si₃N₄, SiO₂ and silicon and L is the thickness of the silicon substrate. The first two factors on the right-hand side of equation (2.3) represent the transmission across the two dielectric layers, while the integral term corresponds to the average charge-collection efficiency (CCE(z)) at the given photon energy. The CCE(z) is defined as the fraction of the total photo-generated charge that is collected by the readout electrodes when the X-ray photon is absorbed at depth z [15]. It can be assumed that the minimum value of the CCE(z) occurs when a photon is absorbed at the surface (z = 0), while it is unity for deep absorption, i.e. when a photon is absorbed within the depleted silicon region.

Regardless of the specific shape of CCE(z), equation (2.3) allows a simple estimation of l_N and l_O . An abrupt decrease in QE is expected at the K absorption edges of nitrogen and oxygen (i.e. at the energies $E_N = 410 \,\text{eV}$ and $E_O = 543 \,\text{eV}$, respectively), due to the discontinuity of the attenuation lengths $\lambda_N(E)$ and $\lambda_O(E)$ (figure 1(a)). Taking into account the Si₃N₄ thickness l_N first, it can be expressed, according to equation (2.3), in terms of the ratio of the QE values at photon energies slightly above and below the nitrogen edge (E_N^+ , E_N^-):

$$l_{\rm N} = \ln \left(\frac{\mathrm{QE}(E_{\rm N}^+)}{\mathrm{QE}(E_{\rm N}^-)} \right) \cdot \frac{\lambda_{\rm N}(E_{\rm N}^-)\lambda_{\rm N}(E_{\rm N}^+)}{\lambda_{\rm N}(E_{\rm N}^+) - \lambda_{\rm N}(E_{\rm N}^-)}$$
(2.4)

The SiO₂ thickness $l_{\rm O}$ can be expressed with an analogous formula, by considering the attenuation length $\lambda_{\rm O}(E)$ instead of $\lambda_{\rm N}(E)$ and by replacing the nitrogen edge $E_{\rm N}$ with the one of oxygen $E_{\rm O}$.

The simplest approach for describing of the CCE(z) is the dead-layer model. It assumes that the charge carriers generated by an X-ray photon absorbed within a thickness l_D from the silicon surface, known as the dead layer, are completely lost due to recombination. Conversly, the charge is completely collected if the photon is absorbed beyond the dead layer. The corresponding expression of $CCE_{dl}(z)$ is:

$$CCE_{dl}(z) = \begin{cases} 0 & \text{if } 0 < z \le l_{D} \\ 1 & \text{if } z > l_{D} \end{cases}$$
 (2.5)

2.3.2 Average gain and multiplication factor

Assuming the same QE for both the iLGAD and the pin-diode coming from the same wafer, the average gain g of an iLGAD diode can be defined as the ratio of the photocurrents from the iLGAD diode and the n^+ -p- p^+ diode:

$$g(E) = \frac{I_{\text{iLGAD}}^{\gamma}(E, \Phi)}{I_{\text{pin}}^{\gamma}(E, \Phi)}.$$
 (2.6)

The multiplication factor *M* is defined as the ratio between the number of electron-hole pairs that are collected at the boundaries of the depletion layer of the sensor and the initial number of electron-hole pairs introduced within it due to the absorption of an X-ray photon. Because of the difference in the impact ionization coefficient between electrons and holes in silicon, the multiplied charge depends on which charge carrier triggers the impact ionization and thus on the position where the X-ray photons are absorbed inside the sensor. In particular:

1. when photons are absorbed in the n^+ implant close to the silicon surface, the multiplication is initiated by holes that drift through the gain layer and towards the p^+ electrode of the sensor;

- 2. if the absorption occurs beyond the gain layer the avalanche is instead triggered by electrons which drift to the n^+ -implant layer;
- 3. if the carrier generation by X-ray photons occurs within the gain layer, the impact ionization is initiated by both primary electrons and holes.

Because the impact ionization coefficient of electrons is larger than that of holes ($\alpha_e > \alpha_h$) [16], the multiplication factor M_h for hole-initiated impact ionization (case 1) is lower than the multiplication factor M_e for avalanches initiated by electrons (case 2). Therefore, the multiplication factor is a function M(z) of the photon absorption depth z. As illustrated in figure 2, $M(z) = M_h$ for $z < t_1$, where t_1 is the onset of the gain layer, $M(z) = M_e$ for $z > t_2$, where t_2 is the end point of the gain layer, and M(z) transitions from M_h to M_e ($M_e > M_h$) for $t_1 < z < t_2$ (case 3).

The average gain g(E) and the multiplication factor M(z) are related according to the equation:

$$g(E) = \frac{\int_0^L CCE(z)M(z)\frac{1}{\lambda_{Si}(E)}\exp\left(-\frac{z}{\lambda_{Si}(E)}\right)dz}{\int_0^L CCE(z)\frac{1}{\lambda_{Si}(E)}\exp\left(-\frac{z}{\lambda_{Si}(E)}\right)dz} \approx \int_0^L M(z)\frac{1}{\lambda_{Si}(E)}\exp\left(-\frac{z}{\lambda_{Si}(E)}\right)dz \quad (2.7)$$

In the last step of equation (2.7), it is assumed that the charge collection is complete (i.e. no charge loss inside the n^+ implant layer, $CCE \approx 1$) and $L \gg \lambda_{\rm Si}(E)$. If these conditions are fulfilled, the average gain corresponds to the mean value of the multiplication factor. A validation of the assumption $CCE \approx 1$ will be given in section 3.1. The average gain of an iLGAD diode is expected to increase with photon energy, because of the increasing probability of photon absorption beyond the gain layer. A saturation of the average gain is expected at large E, when only a negligible fraction of photons is absorbed at $z < t_1$.

Due to the lack of suitable models that can accurately describe the ionization coefficients $\alpha_e(\mathcal{E})$ and $\alpha_h(\mathcal{E})$ in the range of electric field (\mathcal{E}) of LGAD sensors [17], it is difficult to express the multiplication factor analytically to calculate the average gain g(E). Therefore, in this work, the following simple models for the multiplication factor as a function of the depth of absorption have been examined and inserted into equation (2.7). The first model assumes a linear dependence of M(z) on depth z:

$$M_{\text{lin}}(z) = \begin{cases} M_{\text{h}} & \text{if } 0 < z \le t_1\\ \frac{M_{\text{e}} - M_{\text{h}}}{t_2 - t_1} \cdot (z - t_1) + M_{\text{h}} & \text{if } t_1 < z \le t_2\\ M_{\text{e}} & \text{if } t_2 < z \le L \end{cases}$$
(2.8)

The second model considers an exponential transition of M(z) from t_1 to t_2 :

$$M_{\exp}(z) = \begin{cases} M_{\rm h} & \text{if } 0 < z \le t_1 \\ M_{\rm h} \cdot \left(\frac{M_{\rm e}}{M_{\rm h}}\right)^{\frac{z-t_1}{t_2-t_1}} & \text{if } t_1 < z \le t_2 \\ M_{\rm e} & \text{if } t_2 < z \le L \end{cases}$$
 (2.9)

The multiplication factor as a function of depth, and hence the parameters M_e , M_h , t_1 and t_2 , determine the spectral response of hybrid detectors coupled with iLGAD sensors to monochromatic X-rays. The collected signal charge produced by an X-ray photon will depend on its absorption depth, where M_h and M_e respectively define the minimum and maximum charge that can be collected. For

single photon detection at low energies only photons absorbed after the gain layer (electron-triggered multiplication, $M(z) = M_e$) will produce a SNR high enough to be detected. On the other hand, for multiple photon detection, the broad distribution of the multiplication coefficients will result in a reduced accuracy in the conversion of the signal charge into number of photons, since only an average multiplication coefficient can be applied. The relative probability of hole- and electron-triggered multiplication strongly depends on the bounds of the gain layer (t_1 and t_2), since in the soft X-ray energy range most of the X-ray photons are absorbed within the first μm inside the active volume of the sensor below the EW.

3 Results and discussion

3.1 Quantum efficiency vs. photon energy

The QE as a function of photon energy obtained from the measurements using formula 2.2 is shown in figure 5 for all investigated diodes. The QE values for photon energies above 250 eV are larger than 55% for all variations. This result represents a substantial improvement over conventional hard X-ray silicon sensors, which typically show a QE of the order of few % at 250 eV and less than 50% at 800 eV [11].

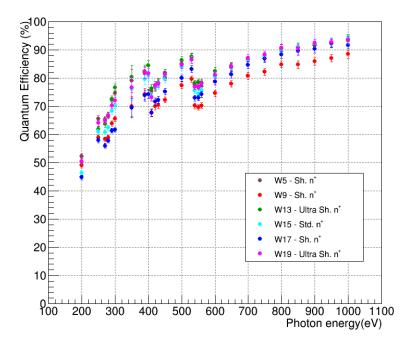


Figure 5. QE as a function of the photon energy for the diodes under test listed in table 1.

The increase in the attenuation length in Si_3N_4 , SiO_2 , and Si with the photon energy, as shown in figure 1, results in a general trend of increasing QE at the higher photon energies. For X-ray photons with higher energies, absorption in the insensitive passivation layer and partial charge collection in the near-surface region as a result of recombination become less probable. In particular, at high energies the attenuation length of X-ray photons is significantly larger than the thicknesses of

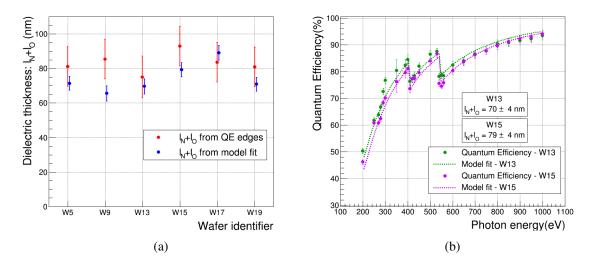


Figure 6. (a) Total thickness of the dielectric layers, $l_N + l_O$, for different diodes. Red markers: evaluation from QE edges (equation (2.4)); blue markers: evaluation from the fit to the QE measurements using formula 2.3 with the dead layer model for CCE(z). (b) Fit of the QE measurements with formula 2.3 with the dead layer model for two representative wafers, W13 and W15.

the dielectric layer and the n^+ implant for the different diodes under test; in this photon energy region the QE values for the different diodes are similar. The diode W9, which shows lower QE values compared to the others at high energies, is an exception. It is assumed that the different behavior is related to a faulty centering of the beam on the diode active area from the beginning: due to the shift of the motor position during the scan, part of the beam hits the border of the diode, which causes a smaller photocurrent and an underestimation of the QE.

As expected, sudden drops in QE are present at the K absorption edges of nitrogen ($E_{\rm N} \sim 410\,{\rm eV}$) and oxygen ($E_{\rm O} \sim 543\,{\rm eV}$), due to the enhancement of the photon losses in the passivation. The thicknesses of the Si₃N₄ and SiO₂ layers, $l_{\rm N}$ and $l_{\rm O}$, are estimated using equation (2.4), where the QE measurements at the photon energies across the edges were used¹. Since the individual thicknesses of the two layers cannot be disclosed, the total thickness $l_{\rm N} + l_{\rm O}$ is shown in figure 6(a) (red markers).

A fit of the QE measurements with expression 2.3 using the dead layer model $CCE_{\rm dl}(z)$ was carried out with fitting parameters l_N , l_O , l_D . For W9, the function in eq. (2.3) was multiplied by another parameter A that represents the fraction of the beam impinging on the active area of the diode (best-fit value $A = 0.904 \pm 0.009$). Figure 6(b) shows QE functions fitted for two representative diodes (W13, W15). In all cases, the best-fit value of l_D is 0 nm, which indicates complete charge collection within the silicon bulk. The deterioration in QE is therefore dominated by photon losses in the inactive dielectric layers of the EW.

In figure 6(a), the sum of $l_{\rm N}+l_{\rm O}$ obtained from the fit to the data from different diodes is plotted in blue. The fit of the data with the function in equation (2.3) yields 5–10 nm lower values of $l_{\rm N}+l_{\rm O}$ compared to the values obtained from the measurement using the QE drops at the K absorption edges. Increasing the fit parameters $l_{\rm N}$ and $l_{\rm O}$ in the QE model to match the amplitudes of the measured QE

 $^{^{1}}$ For $l_{\rm N}$ the QE points used are at $E_{\rm N}^{-}=400\,{\rm eV}$ and $E_{\rm N}^{+}=410\,{\rm eV}$, while for $l_{\rm O}$ at $E_{\rm O}^{-}=530\,{\rm eV}$ and $E_{\rm O}^{+}=550\,{\rm eV}$.

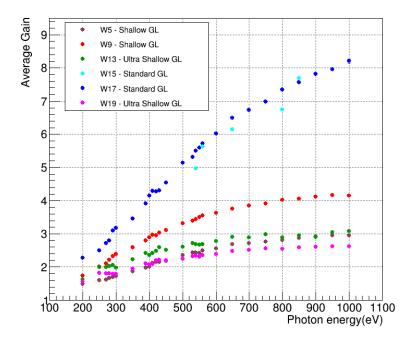


Figure 7. Average gain as a function of the photon energy for the iLGAD diodes listed in table 1. Due to the large fluctuation of current measured for W15 iLGAD, only the results with a relative error below 3.5% are shown for this wafer.

edges results in an overall reduction of the QE and, thus, increases the residuals. The thicknesses obtained for the dielectric layers using the two different methods are consistent within the statistical error.

3.2 Average gain vs. photon energy

Figure 7 displays the average gain of the iLGAD diodes (equation (2.6)). It can be seen that the average gain increases with photon energy, due to the increase of the probability of absorption beyond the gain layer. As discussed in section 2.3.2, this corresponds to electron-initiated avalanches, which are characterized by a higher multiplication factor compared to the hole-initiated ones ($M_e > M_h$).

Because of the thicker gain layer, diodes featuring the standard gain layer design (W15, W17) exhibit higher average gain than diodes with shallow and ultra-shallow designs. The saturation of the average gain at high E is obtained when only a small fraction of the impinging photons is absorbed before or within the gain layer. The saturation is reached for ultra-shallow gain layer designs (W13, W19), while for the standard gain layer (W15, W17) g(E) continues to increase beyond the upper bound of the photon energy in the measurement.

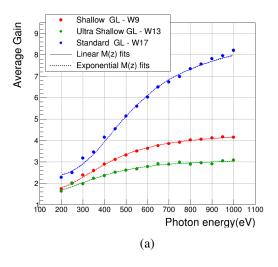
Different average gain values can be observed both between the two iLGADs with shallow gain layer designs (W5 and W9) and between the two with ultra-shallow designs (W19 and W13). These discrepancies can be attributed to lower doses of the *p*-type gain layer in W5 and W19 (table 1). This results in a reduced electric field intensity in the multiplication region, leading to lower impact ionization coefficients of both electrons and holes.

According to the results of section 3.1, the average gain of the iLGADs as a function of energy can be fitted using equation (2.7) and assuming $CCE(z) \approx 1$. From the fit, the multiplication factors of electron-initiated avalanches and hole-initiated avalanches (M_e and M_h) can be extracted, as described in section 3.3.

3.3 Multiplication factor vs. absorption depth

The models $M_{\rm lin}(z)$ and $M_{\rm exp}(z)$ were used in equation (2.7) for the average gain fit, with free parameters $M_{\rm e}$, $M_{\rm h}$, t_2 . The parameter t_1 was fixed at the onset of the depletion layer under full depletion condition. Its value is determined by solving a one-dimensional Poisson equation that takes into account the doping profiles of the n^+ implant and the p-type gain layer and the doping concentration of the substrate. The boundary conditions for the solution are an electric potential of 0 V at the p^+ readout electrode and 300 V at the n^+ contact. The calculated value of t_1 approximates the onset of the region where the electric field is of the order of $\sim 300 \, \text{kV/cm}$ and multiplication occurs.

In figure 8(a) the fits to the average gain data are shown, while the best-fit parameters are reported in table 2. It can be seen that the best-fit functions using the linear and exponential models are almost overlapping, indicating that the measurements of the average gain do not favor one model over the other. Figure 8(b) shows the dependence of the multiplication factor on the depth of absorption for the best-fit linear models and exponential models. The extracted M_e and M_h for both models are consistent. The exponential increase of the multiplication factor in $M_{\rm exp}(z)$ is compensated by a smaller t_2 compared to $M_{\rm lin}(z)$ and yields a similar M_e . In order to study the sensitivity of the best-fit parameters to the fixed value of t_1 , the fits were repeated by increasing or decreasing t_1 by 10% (6.1–11 nm, depending on the iLGAD design). Considering the linear model, the best-fit values of M_h change by 0.80–1.2% (0.012–0.031, depending on the iLGAD), M_e



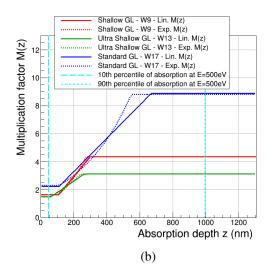


Figure 8. (a) Fits of the average gain g(E) for the iLGADs W9, W13 and W17. The fits were carried out with equation (2.7), using the linear model (equation (2.8)) and the exponential model (equation (2.9)) for M(z). The best-fit functions are similar for both models. (b) The models $M_{\text{lin}}(z)$ (solid lines) and $M_{\text{exp}}(z)$ (dotted lines) with best-fit parameters M_{e} , M_{h} , t_2 for the iLGADs W9 (red), W13 (green), W17 (blue); the 10th and 90th percentiles of the absorption depth for 500 eV photons (cyan).

changes by 0.016–0.14% (0.005–0.012), and t_2 varies by 1.8–3.6% (4.7–14 nm). Similar results are obtained with the exponential model of M(z).

According to the fitted parameters (table 2), the standard gain layer design (W17) provides larger multiplication factors M_e and M_h than the shallow (W9) and ultra-shallow (W13) designs. On the other hand, these factors do not scale with the thickness of the gain layer, $t_2 - t_1$, since they depend on the specific electric field profile within the multiplication region as well. The ratio M_h/M_e is higher in the ultra-shallow (W13) and shallow (W9) designs than the standard design (W17), due to the different dependence of the impact ionization coefficients of electrons (α_e) and holes (α_h) on the electric field [17]. The electric field in the gain layer for W13 and W9 is higher than that for W17, resulting in a higher ratio α_h/α_e and thus higher M_h/M_e .

Table 2. Parameters of the linear and exponential models M_{lin} and M_{exp} from the fits of the average gain for the iLGAD diodes W9, W13 and W17. t_1 is a fixed parameter in the fits. The iLGADs were biased at 300 V and their temperature was stabilized at +20° C.

Device	<i>p</i> -implant	M(z)	$M_{ m e}$	$M_{ m h}$	$t_1(nm)$	$t_2(\text{nm})$	$M_{ m h}/M_{ m e}$
W17	Standard	Lin	8.88 ± 0.03	2.25 ± 0.02	112	670 ± 12	0.253 ± 0.003
VV 17	Medium	Exp	8.81 ± 0.03	2.31 ± 0.02		548 ± 10	0.262 ± 0.003
W9	Shallow	Lin	4.355 ± 0.006	1.64 ± 0.01	107	299 ± 5	0.377 ± 0.003
113	High	Exp	4.352 ± 0.006	1.65 ± 0.01		271 ± 5	0.380 ± 0.003
W13	UltraShallow	Lin	3.12 ± 0.01	1.50 ± 0.04	61	263 ± 18	0.48 ± 0.01
VV 13	High	Exp	3.12 ± 0.01	1.51 ± 0.04		242 ± 17	0.49 ± 0.01

4 Summary and outlook

The adaptation of the hybrid detector technology for soft X-ray detection involves the development of iLGAD sensors with an optimized EW to improve both the quantum efficiency (QE) and the signal-to-noise ratio (SNR). Variations of diodes of this type with and without gain, fabricated by FBK, have been characterized using soft X-rays in the photon energy range between 200 eV and 1 keV at the SIM beamline of the SLS synchrotron.

The QE is in the range 55–67% at 250 eV for all process variations, a significant improvement compared to conventional planar sensors. Additionally, the QE increases with photon energy due to the increase of the attenuation length, with discontinuities at the K absorption edges of nitrogen and oxygen, because of the use of SiO_2 and Si_3N_4 dielectric layers in the EW. The dependence of the QE on photon energy can be described with a model that considers photon absorption in the dielectric layers and carrier recombination in a dead layer of silicon close to the silicon surface. With this model, we showed that the dominant cause of QE degradation is photon loss in the dielectric layers (total thickness ≈ 70 –90 nm), while the charge collection is almost 100% inside the silicon sensor (CCE ≈ 1). This study suggests that the thicknesses of the dielectric layers need to be reduced, without sacrificing their uniformity over a large area, for further improvement of the QE.

For all iLGAD diodes studied, the average gain g(E) increases with photon energy. This behavior is attributed to the increased likelihood of electron-initiated multiplication (corresponding to absorption beyond the gain layer) at higher photon energies. The standard gain layer design yields a higher

average gain with respect to the shallow and ultra-shallow ones. In addition, iLGADs with a higher dose in the p-type gain layer (with shallow or ultra-shallow designs) exhibit a higher average gain.

Two empirical models were introduced for g(E), which involve a linear and an exponential transition of the multiplication factor M(z) as a function of the absorption depth z from the value M_h (corresponding to $z < t_1$ i.e. absorption before the gain layer and hole-initiated avalanches) to $M_{\rm e}$ (corresponding to $z > t_2$, i.e. absorption beyond the gain layer and electron-initiated avalanches). Both models describe the measured g(E) well and the fitted values of M_h and M_e turn out to be independent of the model chosen. The standard gain layer design produces larger M_h and M_e , potentially leading to an higher SNR for single photon detection using hybrid pixel detectors in comparison to the shallower designs. On the other hand, the shallow and ultra-shallow designs take advantage from a higher probability of electron-triggered multiplication (lower t_2 value) at a given photon energy. This property may provide a higher photon detection efficiency at low energy, if the SNR associated with electron-triggered multiplication is sufficient to achieve single photon detection. In addition, the ratio M_h/M_e is higher for the shallow and ultra-shallow gain layer designs, because of the higher the electric field intensity in the multiplication region, compared to the standard one. A higher ratio M_h/M_e could be an advantage for the operation of both single-photon counting and charge-integrating hybrid detectors, since it results in more similar amplitudes of the output signals due to hole- and electron-triggered multiplication of the charge produced by an X-ray photon.

The fitting procedure presented in this work provides a way to estimate both M_e and M_h from the measurements of the average gain of the iLGADs. It is valuable for future investigation of their dependence on electric field and temperature. Further studies of M(z) in iLGAD sensors will entail measuring single photon spectra of hybrid pixel charge-integrating detectors (e.g. MÖNCH [5] and JUNGFRAU [3]) coupled with the same iLGADs used in this work. The extracted parameters of the linear and exponential models for M(z) will serve as a basis for simulations of these spectra. By comparing the simulation results with experimental data, further insight will be also obtained into the transition of M(z) between M_h and M_e .

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